

**Method Article****Rapid chemical bath deposition and optical property of  
CuS films using sodium ethylenediamine tetraacetate as  
chelating agent**

**Abstract:** CuS films were fabricated by a chemical bath deposition process in an aqueous solution of cupric acetate, thiourea, and sodium ethylenediamine tetraacetate (EDTA-2Na) with different contents at 50 °C. The films were characterized by X-ray diffraction, atomic force microscopy, ultraviolet-visible spectrophotometry and photoluminescence spectrophotometry. The effects of EDTA-2Na content and deposition time on the thickness, transmittance, band gap energy, and photoluminescence of the films were investigated. Molar ratio of EDTA-2Na/Cu<sup>2+</sup>=1.0 led to a maximum deposition rate. The band gap energy of the films ranged from 2.59–2.92 eV. The films showed broad emission centered at ~443 nm. The intensity of the emission decreased with increasing deposition time.

**Keywords:** CuS; film; deposition; transmittance, bandgap; luminescence

**1. Introduction**

Copper sulfides (Cu<sub>x</sub>S, x = 1–2) are significant binary compounds that attract much attention due to their wide range of applications in optical and electrical devices, such as photo thermal conversion, microwave shielding coatings, solar control coatings, dye-sensitized solar cells, potential nanometer-scale switch, cathode materials in lithium rechargeable batteries and some chemical sensors [1-6]. Additionally, it has

23 recently been reported that CuS can transform into a superconductor below the 1.6 K  
24 because of its metallic conduction property [7].

25 Even though physical techniques are very suitable for synthesizing uniform and high  
26 quality films, they are also very expensive and consume large amounts of energy.  
27 Additionally, they inevitably destroy the film morphology at high temperatures above  
28 500 °C [8]. On the other hand, chemical methods are economical and suitable for  
29 maintaining desired structures. Thus, to prepare CuS thin films with desired shapes  
30 and structures, many different chemical methods have been developed, such as  
31 liquid–liquid interface reaction [9], chemical vapor deposition (CVD) [10], chemical  
32 vapor reaction (CVR) [11], electrochemical method [12], successive ionic layer  
33 adsorption and reaction (SILAR) [13], atomic layer deposition (ALD) [14], spray-ion  
34 layer gas reaction (ILGAR) [15], chemical bath deposition (CBD) [16–18], etc.  
35 Among them, CBD tends to be a better method to deposit CuS films with particular  
36 shape, orientation and thickness because of the advantages of having a variety of  
37 substrates (insulators, semiconductors and metals) for deposition, a large surface area,  
38 simplicity, and lower cost. However, CBD generally contains Cu salt, triethanolamine,  
39 thiourea, and large amount of sodium hydroxide and/or ammonia for most case. With  
40 such solution the deposition usually required a significant long time to achieve the  
41 appropriate film thickness. To enhance the deposition rate, microwave [19] and  
42 thermal [20–22] assistances has been introduced to the CBD. In addition, small  
43 amount of other chemical bath solution composition, such as, aqueous solutions of  
44 copper sulphide and thioacetamide mixed together with chelate agent of acetic acid

45 [18], aqueous solution of copper nitrate, concentrated ethylene glycole ( $C_2H_6O_2$ ),  
46 thioacetamide with different pH [22], etc., has been used for CBD growth of the CuS  
47 film.

48 In this paper, we present (i) fast fabrication of uniform CuS films with a simple  
49 CBD; (ii) the effects of chemical bath solution composition and deposition time on  
50 the thickness and the optical properties of the deposited films.

51

## 52 **2. Experiment method**

### 53 2.1. Deposition of the film

54 The starting materials used were all analytic grade chemicals without any further  
55 processing. The bath solution was composed of 0.01 mole cupric acetate  
56 ( $Cu(CH_3COO)_2 \cdot H_2O$ ), 0.015 mole thiourea ( $SC(NH_2)_2$ ), 80 ml deionized water  
57 and different amounts of disodium ethylenediamine tetraacetate (EDTA-2Na,  
58  $C_{10}H_{14}N_2Na_2O_3$ ) (0.005, 0.01, and 0.015 mole). The solution was homogenized by  
59 magnetic stirring for 10 min in a beaker at room temperature. Commercial glass slides  
60 were used as a substrate and were ultrasonically cleaned in acetone and methanol  
61 respectively for 0.5 h. At room temperature ( $\sim 20^\circ C$ ), the film deposited for 7.5 h was  
62 grey white, indicating it can not be CuS crystal phase. Therefore the deposition  
63 temperature of  $50^\circ C$  was selected. After the solutions were put into a furnace, the  
64 furnace was heated from room temperature to  $50^\circ C$  and maintained for 10 min to  
65 make actually solution temperature reach to  $50^\circ C$ . Three glass substrates were then  
66 put vertically into each of the three solutions. After 0.5 h, 1.0 h, and 1.5 h of

67 deposition, the films were successively taken out from the solutions, and rinsed with  
 68 deionized water to remove the residual bath solution and loosely adhered CuS  
 69 particles on the films and then naturally dried in air.

## 70 2.2. Characterization of the films

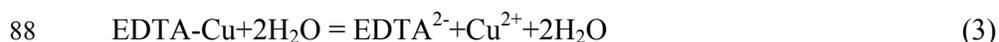
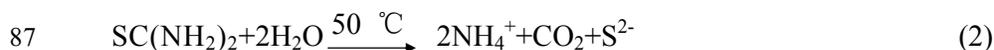
71 The phase and structure of the deposited CuS thin films were identified at room  
 72 temperature using an X-Ray diffractometer (XRD,  $\text{CuK}_{\alpha 1}$ ,  $\lambda=0.15406\text{nm}$ , Model No:  
 73 D/Max-2200PC, Rigaku, Japan). The morphology and texture of the films were  
 74 analyzed using atomic force microscopy (AFM, Model No: SPI3800N, NSK,  
 75 Japan). The transmittance spectra of the films was determined with an  
 76 ultraviolet-visible spectrophotometer (Model No: 752N, Feilo, China). The  
 77 photoluminescence spectrum of the films was measured on a photoluminescence  
 78 spectrophotometer (Model no: F-4600, Hitachi, Japan).

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## 80 3. Results and discussion

81 Figure 1 shows the XRD patterns of the CuS films. The main phase of the films was  
 82 hexagonal covellite CuS (JCPDS: 05-0464). The peak at  $2\theta\sim 18^\circ$  was identified as  
 83 yarrowite  $\text{Cu}_2\text{S}$  (JCPDS: 36-0379) which decreased with the increase in deposition  
 84 time and disappeared when the deposition time was 1.5 h.

85 The chemical process for the CuS film formation can be given by:





90 CuS has a very small solubility constant ( $K_{sp}=1.27 \times 10^{-36}$ ) and so is easily formed  
91 from the  $\text{Cu}^{2+}$  and  $\text{S}^{2-}$  ions in the aqueous solution. However, the  $\text{Cu}^{2+}$  can be chelated  
92 with EDTA, so the release rate of  $\text{Cu}^{2+}$  from the chelate compound EDTA-Cu could  
93 control the formation rate of the CuS.

94 Figure 2 shows AFM micrographs of the CuS films deposited with EDTA-2Na/ $\text{Cu}^{2+}$   
95 =1.0. The films were composed of uniform particles. With increasing deposition time,  
96 the average particle size increased and particle morphology tended to be rodlike. The  
97 thickness of the films determined from AFM analysis are shown in Fig. 3. The  
98 thickness increased as increase in deposition time. The maximal deposition rate was  
99 achieved when the moderate molar ratio of EDTA-2Na/ $\text{Cu}^{2+}$ =1.0. At lower molar ratio  
100 of EDTA-2Na/ $\text{Cu}^{2+}$  = 0.5, excessive  $\text{Cu}^{2+}$  anions in the solution led to the formation  
101 of CuS particle too fast and undesirably large particles formed. Particles of such size  
102 have difficulty tightly adhering on the film, which resulted in slower deposition rate  
103 of the film. At higher molar ratio of EDTA-2Na/ $\text{Cu}^{2+}$  =1.5, the release of  $\text{Cu}^{2+}$  anion  
104 from the EDTA-Cu was slower, and so the deposition rate was lower.

105 Figure 4 shows the transmittance spectra of the CuS films fabricated with different  
106 EDTA-2Na proportions. The transmittance of the films decreases with increasing the  
107 deposition time for each EDTA-2Na proportion. The films fabricated with  
108 EDTA-2Na/ $\text{Cu}^{2+}$ =1.0 had the greatest decrease in transmittance with deposition time,  
109 which was consistent with the increases of the film thicknesses as shown in Fig. 3.  
110 However, the decrease in transmission with the deposition time does not seem to scale

111 with the film thickness. This could be mainly attributed to higher transmittance of  
112  $\text{Cu}_2\text{S}$  than  $\text{CuS}$  [23, 24].

113  $\text{CuS}$  is considered as a direct band gap semiconductor, so the relationship between  
114 the absorption coefficient ( $\alpha$ ) near the absorption edge and the optical band gap  
115 energy ( $E_g$ ) for direct band transitions obeys the following formula [25]:

$$116 \quad (\alpha hv)^2 = C(hv - E_g)$$

117 where  $hv$  is photon energy and  $C$  is a constant. We assume

$$118 \quad \alpha = \left(\frac{1}{d}\right) \left(\ln \frac{1}{T}\right)$$

119 where  $T$  is the transmittance and  $d$  is the film thickness. Fig. 5 shows the graphs of  
120  $(\alpha hv)^2$  vs. photon energy,  $hv$ , for the  $\text{CuS}$  thin films. The linear dependence of  $(\alpha hv)^2$   
121 on  $hv$  at higher photon energies indicates that the  $\text{CuS}$  films are essentially direct-  
122 transition-type semiconductors. The straight-line portion of the curve, as extrapolated  
123 to zero, gives the optical band gap energy  $E_g$ . The estimated  $E_g$  was in range of  
124 2.59–2.92 eV which decreased with increasing deposition time (Fig. 6). For a  
125 comparison, Gadave and Lokhande [26] found optical band gap energy of 2.40 eV for  
126 CBD deposited  $\text{Cu}_x\text{S}$  film from aqueous medium. Sartale and Lokhande [27] reported  
127 optical band gap energy of 2.36 eV of SILAR deposited  $\text{Cu}_x\text{S}$  from nonaqueous  
128 medium. Puspitasari et al [18] reported optical band gap energy of 2.8 eV for the  
129  $\text{CuS}$  film prepared by a simple chemical method. The variation of the band gap  
130 energy with deposition time has been reported by some researcher [19]. They reported  
131 that the band gap energy of the  $\text{CuS}$  film decreased and then increased with  
132 microwave-assisted deposition time. The detailed reason for this variation is unclear  
133 at present. We believe that the different phases, crystal orientation and crystallinities

134 are responsible for this variation.

135 Figure 7 shows the photoluminescence spectra of the CuS films deposited at the  
136 EDTA-2Na/Cu<sup>2+</sup>=1.0. They were measured at excitation wavelength of 350 nm. The  
137 emission peaks were centered at ~443 nm. This value is different from that of CuS  
138 nanorods (420, 450 and 462 nm [28], and 515 nm [29]) and nanoplates (339 nm [30]).  
139 The emission peaks of ~443 nm corresponded to photon energy of ~2.79 eV that is  
140 closed to the band gap energy estimated from the transmittance spectra. This indicated  
141 that the emission could come from a near band gap emission. The decrease in the  
142 intensity of the emission with the deposition time can be also observed in Fig. 7. Roy  
143 et al [29] ascribed the decrease in the luminescence intensity of their powders to the  
144 increase of particle size due to the decrease in the content of surface oxygen vacancy  
145 and defect with increasing the size of materials. In fact, their result indicated that the  
146 decrease in the emission intensity was also accompanied by morphology change from  
147 nanoparticles to nanorods. Therefore, the decrease in the emission intensity of the  
148 CuS films could be associated with the change in the morphology and the crystal  
149 orientation (Fig. 1) and the decrease in the particle size of the films.

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#### 151 **4. Conclusion**

152 CuS semiconductor films were fabricated on a glass substrate in a simple chemical  
153 bath solution containing EDTA-2Na as the chelating agent. Thermal assistance was  
154 used to accelerate deposition. The effect of EDTA-2Na proportion on the deposition  
155 rate and optical properties of the films were investigated, and it was found that the  
156 molar ratio of EDTA-2Na/Cu<sup>2+</sup>=1.0 was most effective. At this proportion, the film

157 thickness reaches ~160 nm in a relative short deposition time of 1.5 h. The optical  
158 band gap energy of the films is in range of 2.59–2.92 eV. The films showed the  
159 emission peaks entered at ~443 nm. The optical properties have some dependences of  
160 microstructure including morphology, particle size and crystal orientation. We believe  
161 this simple chemical bath deposition technique can be further extended to the  
162 fabrication of other semiconductor films.

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## 165 **References**

166 [1] I. Grozdanov, M. Najdoski, Optical and electrical properties of copper sulfide  
167 films of variable composition, *J. Solid State Chem.* 114 (1995) 469–475.

168 [2] L. Huang, P.K. Nair, M.T.S. Nair, R.A. Zingaro, E.A. Meyers, Interfacial  
169 diffusion of metal atoms during air annealing of chemically deposited ZnS-CuS  
170 and PbS-CuS thin films, *J. Electrochem. Soc.* 141 (1994) 2536–2540.

171 [3] H. Lee, S.W. Yoon, E.J. Kim, J. Park, In-Situ Growth of copper sulfide  
172 nanocrystals on multiwalled carbon nanotubes and their application as novel solar  
173 cell and amperometric glucose sensor materials, *Nano Lett.* 7 (2007) 778–784.

174 [4] T. Sakamoto, H. Sunamura, H. Kawaura, T. Hasegawa, T. Nakayama, M. Aono,  
175 Nanometer-scale switches using copper sulfide, *Appl. Phys. Lett.* 82 (2003)  
176 3032–3034.

177 [5] J. S. Chung, H. J. Sohn, Electrochemical behaviors of CuS as a cathode material  
178 for lithium secondary batteries, *J. Power Sources* 108 (2002) 226–231.

179 [6] A.A. Sagade, R. Sharma, Copper sulphide ( $\text{Cu}_x\text{S}$ ) as an ammonia gas sensor

- 180 working at room temperature, *Sens. Actuators B* 133 (2008) 135–143.
- 181 [7] K. Tezuka, W. C. Sheets, R. Kurihara, Y. J. Shan, H. Imoto, T. J. Marks, K. R.  
182 Poepfelmeier, Synthesis of covellite (CuS) from the elements, *Solid State Sci.* 9  
183 (2007) 95–99.
- 184 [8] A. Bauger, J. C. Mutin, J. C. Niepce, Synthesis reaction of metatitanate BaTiO<sub>3</sub>, *J.*  
185 *Mater. Sci.* 18 (1983) 3041–3045.
- 186 [9] C. N. R. Rao, K. P. Kalyanikutty, The Liquid–liquid interface as a medium to  
187 generate nanocrystalline films of inorganic materials, *Acc. Chem. Res.* 41 (2008)  
188 489–499.
- 189 [10] M. Kemmler, M. Lazell, P. O. Brien, D. J. Otway, J.-H. Park, J. R. Walsh, The  
190 growth of thin films of copper chalcogenide films by MOCVD and AACVD  
191 using novel single-molecule precursors, *J. Mater. Sci. Electron.* 13 (2002)  
192 531–535.
- 193 [11] K. J. Wang, G. D. Li, J. X. Li, Q. Wang, and J. S. Chen, Formation of  
194 single-crystalline CuS nanoplates vertically standing on flat substrate, *Cryst.*  
195 *Growth Des.* 7 (2007) 2265–2267.
- 196 [12] T. Kuzuya, K. Itoh, M. Ichidate, T. Wakamatsu, Y. Fukunaka, K. Sumiyama,  
197 Facile synthesis of nearly monodispersed copper sulfide nanocrystals,  
198 *Electrochem. Acta* 53 (2007) 213–217.,
- 199 [13] H. M. Pathan, J. D. Desai, C. D. Lokhande, Modified chemical deposition and  
200 physico-chemical properties of copper sulphide (Cu<sub>2</sub>S) thin films, *Appl. Surf. Sci.*  
201 202 (2002) 47–56.

- 202 [14] J. Johansson, J. Kostamo, M. Karppinen, L. Niinistö , Growth of conductive  
203 copper sulfide thin films by atomic layer deposition, *J. Mater. Chem.* 12 (2002)  
204 1022–1026.
- 205 [15] C.H. Fischer, H.J. Muffler, M. Bar, T. Kropp, A. Schonmann, S. Fiechter, G.  
206 Barbar and M.C. Lux-Steiner, Spray-ion layer gas reaction (ILGAR)—a novel  
207 low-cost process for the deposition of chalcopyrite layers up to the micrometer  
208 range for photovoltaic applications, *J. Phys. Chem. B* 107 (2003) 7516–7521.
- 209 [16] K. D. Yuan, J. J. Wu, M. L. Liu, L. L. Zhang, F. F. Xu, L. D. Chen, F. Q. Huang,  
210 Fabrication and microstructure of *p*-type transparent conducting CuS thin film and  
211 its application in dye-sensitized solar cell, *Appl. Phys.Lett.* 93 (2008) 132106.
- 212 [17] P. K. Nair, V. M. Garcia, O. Gomez-Daza, M. T. S. Nair, High thin-film yield  
213 achieved at small substrate separation in chemical bath deposition of  
214 semiconductor thin films, *Semicond. Sci. Technol.* 16(2001) 855–863.
- 215 [18] I. Puspitasari, T. P. Gujar, K. D. Jung, O. S. Joo, Simple chemical preparation of  
216 CuS nanowhiskers, *Mater. Sci. Eng. B* 140 (2007) 199–202.
- 217 [19] Mudi Xin, Kunwei Li, Hao Wang, Synthesis of CuS thin films by microwave  
218 assisted chemical bath deposition, *Appl. Surf. Sci.* 256 (2009) 1436–1442
- 219 [20] E. Fatas, T. Garcia, C. Montemayor, A. Medina, E.Garcia Camarero, F. Arjona,  
220 Formation of  $Cu_xS$  thin films through a chemical bath deposition process, *Mater.*  
221 *Chem. Phys.* 12 (1985) 121-128
- 222 [21] N. Mukherjee, A. Sinha, G. G. Khan, D. Chandra, A. Bhaumik, A. Mondal, A  
223 study on the structural and mechanical properties of nanocrystalline CuS thin

- 224 films grown by chemical bath deposition technique, *Mater. Res. Bull.* 46 (2011)  
225 6–11
- 226 [22] E. Güneri, A. Kariper, Optical properties of amorphous CuS thin films deposited  
227 chemically at different pH values, *J. Alloys Compd.* 516 (2012) 20-26
- 228 [23] I. Grozdanov, C.K. Barlingay, S.K. Dey, M. Ristov, M. Najdoski, Experimental  
229 study of the copper thiosulfate system with respect to thin-film deposition, *Thin*  
230 *Solid Films* 250 (1994) 67–71.
- 231 [24] Yung-Tang Nien, In-Gann Chen, Rapid thermal annealing of chemical bath-  
232 deposited  $Cu_xS$  films and their characterization, *J. Alloys Compd.* 471 (2009)  
233 553–556
- 234 [25]. N. Serpone, D. Lawless, R. Khairutdinov, Size effects on the photophysical  
235 properties of colloidal anatase  $TiO_2$  particles: Size quantization or direct  
236 transitions in this indirect semiconductor, *J. Phys. Chem.* 99(1995) 16646-16654
- 237 [26] K. M. Gadave, C. D. Lokhande, Formation of  $Cu_xS$  films through a chemical  
238 bath deposition process, *Thin Solid Films* 229 (1993) 1.
- 239 [27] S. D. Sartale, C. D. Lokhande, Studies on large area ( $\sim 50\text{ cm}^2$ )  $MoS_2$  thin films  
240 deposited using successive ionic layer adsorption and reaction (SILAR) method,  
241 *Mater. Chem. Phys.* 65 (2000) 63
- 242 [28] P. Roy and S. K. Srivastava, Hydrothermal growth of CuS nanowires from Cu-  
243 dithiooxamide, a novel single-source precursor, *Cryst. Growth Des.* 6 (2006)  
244 1921–1926

245 [29] P. Roy, S. K. Srivastava, Low-temperature synthesis of CuS nanorods by simple  
 246 wet chemical method, Mater. Lett. 61 (2007) 1693–1697

247 [30] J. Zhang, Z.-K. Zhang, Hydrothermal synthesis and optical properties of CuS  
 248 nanoplates, Mater. Lett. 62 (2008) 2279–2281

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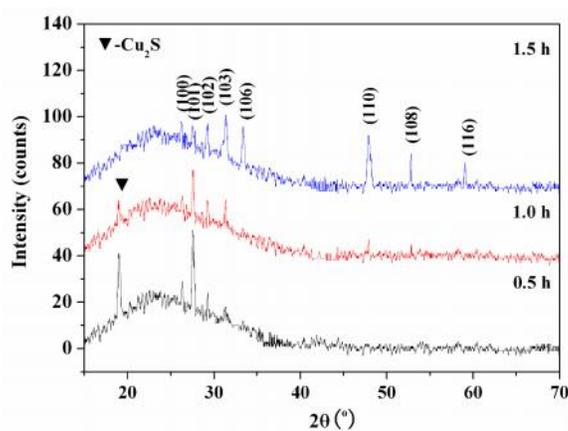
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259 **Figures:**

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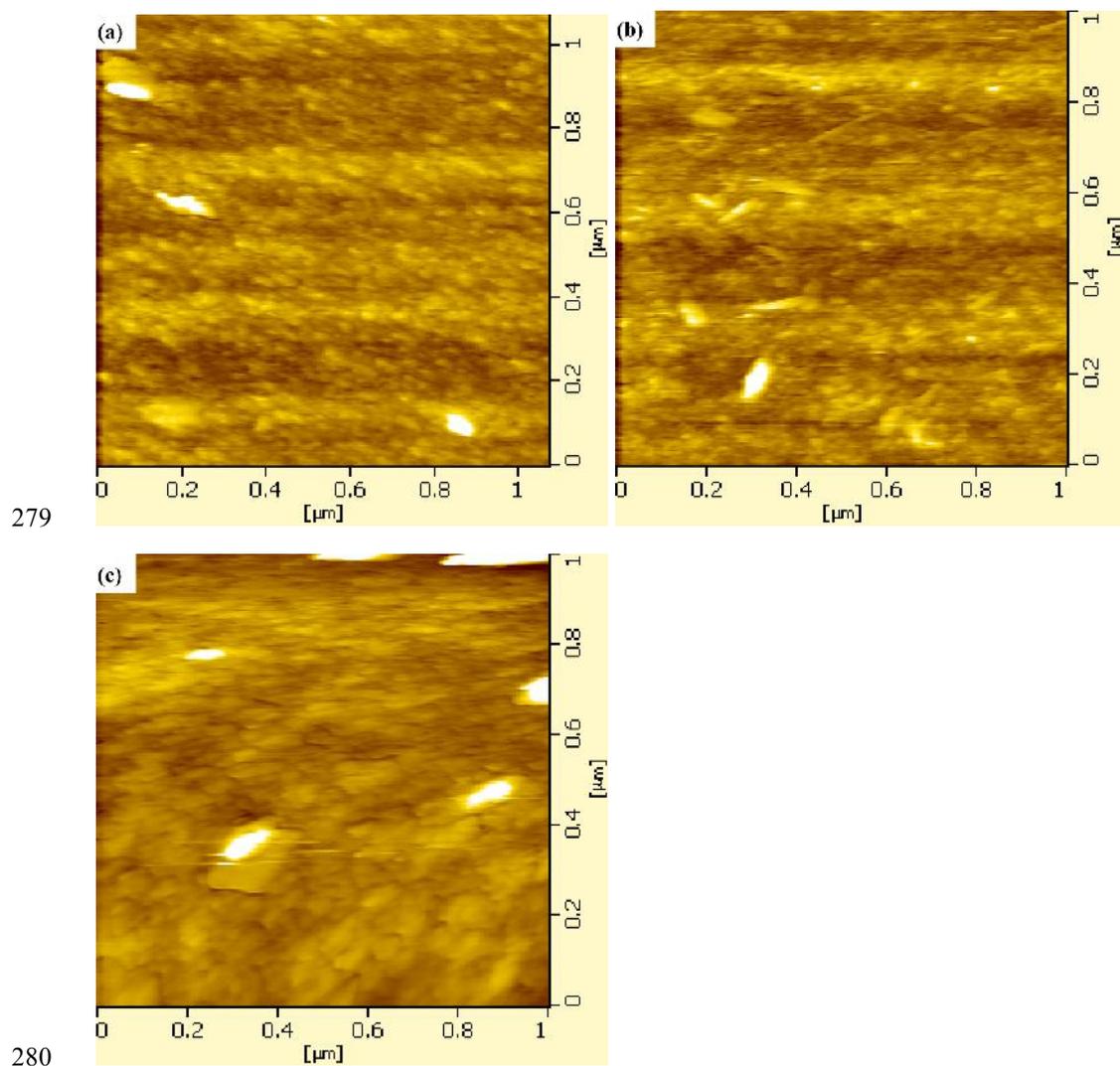


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262 Figure 1. XRD patterns of the CuS films fabricated at the molar ratio of

263 EDTA-2Na: Cu<sup>2+</sup> = 1.0

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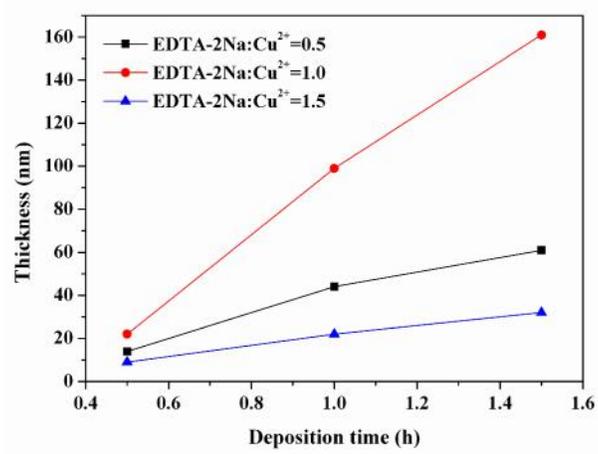
281 Figure 2. AFM micrographs of the CuS films deposited at molar ratio of

282 EDTA-2Na:Cu<sup>2+</sup>=1.0 for deposition time of (a) 0.5 h, (b) 1.0 h, and (c)

283 1.5 h

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288 Figure 3 . Film thickness vs. deposition time

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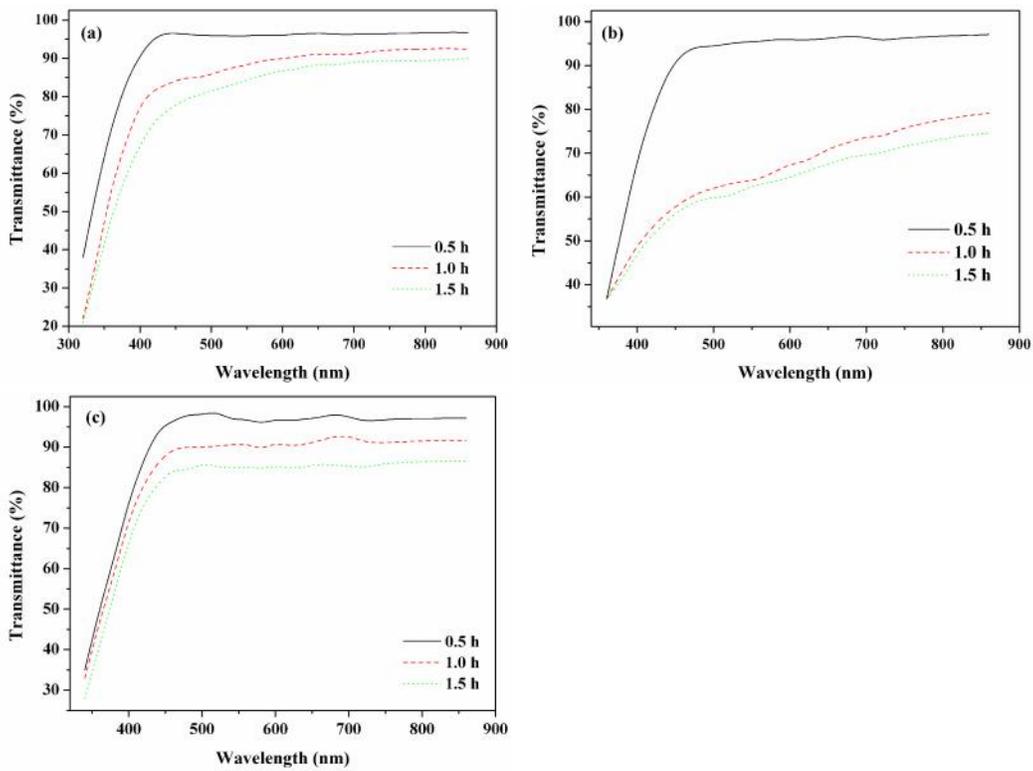
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312 Figure 4. Transmittance spectra of the CuS films deposited at molar ratio of

313 EDTA-2Na: Cu<sup>2+</sup> = (a) 0.5, (b) 1.0, and (c) 1.5.

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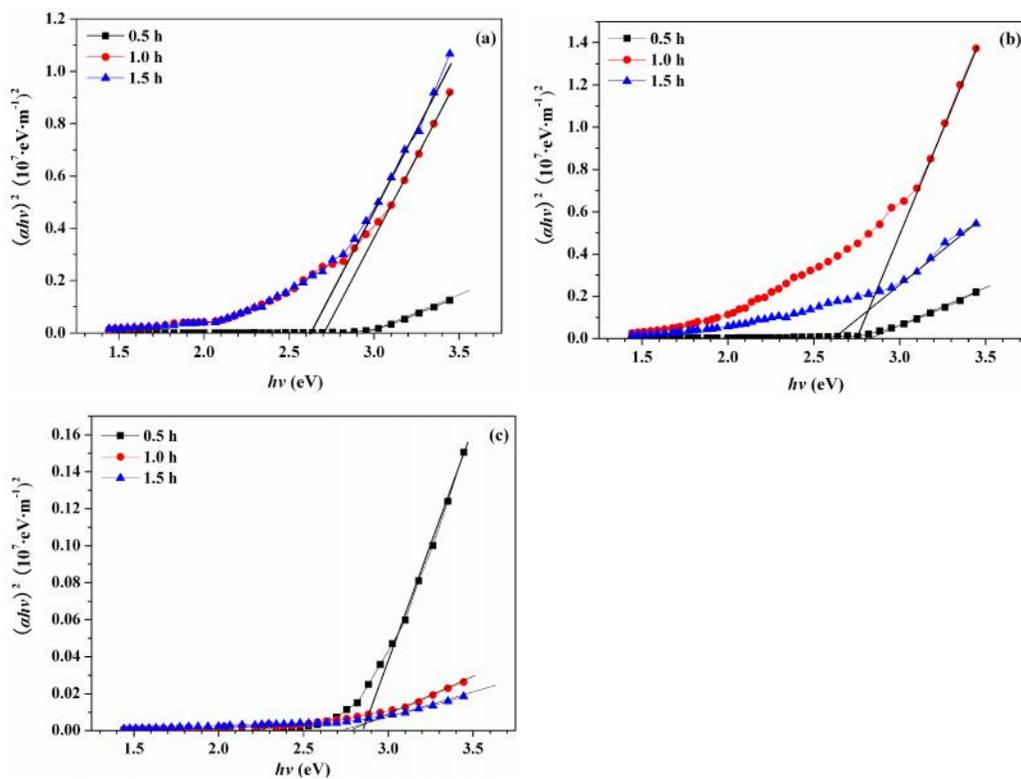
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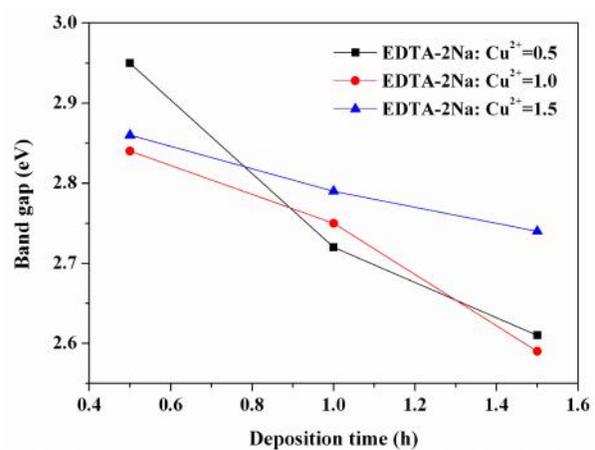
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322 Figure 5. Plots of  $(ah\nu)^2$  vs.  $h\nu$  of the CuS films deposited at molar ratio of

323 EDTA-2Na:  $\text{Cu}^{2+}$ =(a)0.5, (b)1.0, and (c)1.5

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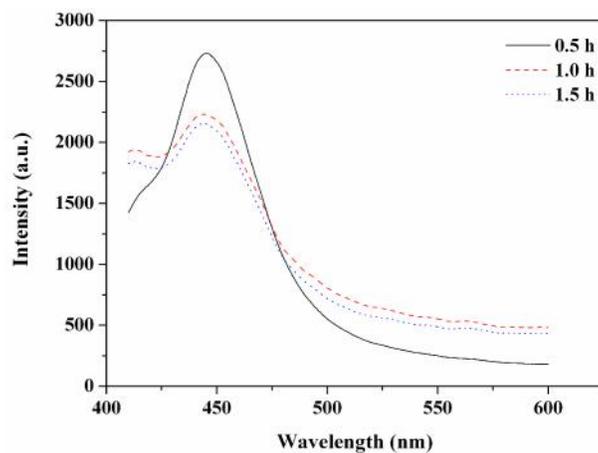
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327 Figure 6 . Optical band gap of the films deposited at at molar ratio of

328 EDTA-2Na:  $\text{Cu}^{2+}$  as a function of the deposition time

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333 Figure 7. Photoluminescence spectra of the CuS films deposited at EDTA-2Na:

334  $\text{Cu}^{2+}=1.0$  and excited at excitation wavelength of 350 nm.

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